

Phase III Compared experiments regarding the laser pyrolysis synthesis of the TiO₂ nanoparticles covered with siloxane polymers: complex characterization.

In the literature there are few studies of the effect of polymeric dispersants adsorbed on the surface of TiO₂ nanoparticles and especially those of change using polycarbosiloxanic polymers are missing. Instead, were tested on foil packaging plastic food ("food packaging") in which dispersed TiO₂ nanoparticles inhibit / destroy bacteria by photochemical sterilization

1.1. Synthesis by laser pyrolysis of TiO₂ nanoparticles coated by carbosilanic polymer (HDMSO precursor)

As we have reported in previous phases, the laser pyrolysis experimental installation used by us, precursors and other auxiliary gas (confinement, dilution, oxidant sensitivity) are inserted through an injector consists of three cylindrical tubes (Figure 1) located about 7 mm below the infrared beam emitted by the CO₂ laser, narrowed in this area to a diameter of 5 mm with a ZnSe lens. Another stream of gas introduced into the reaction chamber is the protection of windows, each 850 sccm argon, serving to prevent any particles to attach to them, and which could result in local absorption of laser beam to create incandescent points resulted in the destruction of their windows.

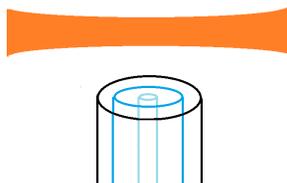


Fig.1 Schematic representation of the injector and the laser beam (in reality invisible to the naked eye)

At this stage, we performed several experiments, nine of them are presented here, two of which were used as references (TLOP01 and TLOP04), meaning that no silicon precursor was introduced at all. And gas flow rates introduced are summarized in Table 1

Exp.	Gas flow by nozzle										
	Central [sccm]				Intermedaite [sccm]					External [sccm]	
	Air>	C ₂ H ₄ >	Add.	C ₂ H ₄ >	Air>	C ₂ H ₄ >	Add.	C ₂ H ₄ >	Ar	C ₂ H ₄	Ar

	TiCl ₄	TiCl ₄	Air	[(CH ₃) ₃ Si] ₂ O	TiCl ₄	TiCl ₄	Air	[(CH ₃) ₃ Si] ₂ O			
TLOP9				20	200	60	400				1700
TLOP10				20	100	30	200				1500
TLOP11			40	20	110	20	450				1700
TLOP04	200	60					200			20	1700
TLOP01	200	60							1700		
TLOP5	200	60						20	200		1700
TLOP6	200	60					200	20			1700
TLOP7	200	60					300	20			1700
TLOP8	200	60					400	20			1700

Table 1. Presentation of the gas / vapor flows introduced into the experiments

The difference between the two experiments is that the reference to TLOP 01 were used only two of the three nozzles, argon confinement being introduced through intermediate nozzle, while in the TLOP 04 20 sccm of ethylene and another of 200 sccm of air was introduced through the intermediate, analogous to the one used in the experiment TLOP6.

Note that for entraining the TiCl₄ vapors we used a mixture of air and ethylene. Also ethylene was used for bubbling and training hexamethyldisiloxane vapors [(CH₃)₃Si]₂O, denoted HMDSO. Pressure of 550 mbar was used to facilitate the involvement of as much quantity of titanium and silicon precursors at room temperature from those two bubblers. Maintaining constant pressure in the reaction chamber was achieved by equalizing the input streams at the output with an electro valve fitted with a clamshell that can adjust the gas flow exiting the filter chamber, located above the reaction chamber.

The laser power was 350 W, which means a laser power density of about 1780 W/cm². This amount of power density was chosen to simultaneously fulfill two conditions: first must be high enough to initiate a stable laser pyrolysis flame, on the other side does not have to be too high because it would enhance the decomposition of ethylene to carbon and the formation of titanium dioxide rutile phase in a greater proportion to its most active phase, anatase.

In the first set of experiments, TLOP 5 ... 8, the Ti precursor is introduced through the central nozzle and Si precursor through the intermediate nozzle, varying the airflow only additional intermediate nozzle from zero to 400 sccm (in TLOP5 instead of air introduced an even flow of Ar to keep the same speed gas from TLOP6 experiment, and to limit oxidation of HMDSO in order to obtain polymers mainly carbosiloxanes instead of silica).

In the second set of experiments TLOP 9 ... 11, the position of the precursors flows was reversed, so that HMDSO is introduced through the central nozzle and TiCl₄ through the intermediate nozzle reducing both the ethylene flow and that the air (vapor carriers of TiCl₄), since ethylene bearing HMDSO flow was maintained constant, the same as in the first series of experiments, 20

sccm. The last two experiments, although the total amount of TiCl_4 involved was the same, we used 100 sccm air and 30 sccm C_2H_4 (TLOP10) or 120 sccm air and 20 sccm C_2H_4 (TLOP11) ethylene flow to TLOP11 being lowest (and the air is greater) was intended to result in less carbon. Thus, in the TLOP11 experiment was much larger amount of air, leading to a total of $40 + 110 + 450 = 700$ sccm to a total of $100 + 200 = 300$ sccm for TLOP10 and finally to a total of $200 + 400 = 600$ sccm for TLOP9 experiment. Due to the lower Ti precursor amount introduced in the last two experiments (TLOP 10 and 11), it is expected that the atomic percent of Ti in the corresponding powders to be lower. Other difference between the two series of experiments arose from the difference between the cross-sectional area of inner nozzle and the concentric nozzle. Thus at equal flow through those nozzles, the central flow velocity will be 10 times greater and the time of the species passing through the laser beam will be considerably lower



Fig.2. Photographies of laser oxidative pyrolysis flames from TiCl_4 , HMDSO and air mixtures

One can see from the fig.2 that the laser oxidative flames extend upstream, outside from the laser beam. This behavior is due to the ethylene burning in air. TLOP-7 and TLOP11 flames also extend downward (again outside from IR laser beam) very near to the glass injector end due to the higher amount of oxidant available.

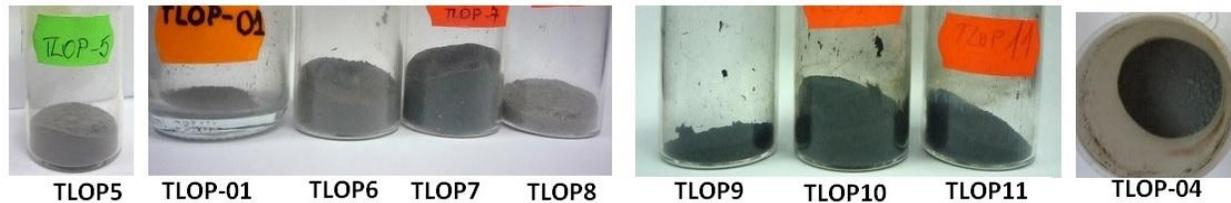


Fig.3. Photographies of the as-obtained raw powders

The grey color of powders is due to their carbon content provided by ethylene decomposition in the presence of insufficient amount of oxidant. The dark blue color can be explained by the supplementary presence of Ti^{3+} ions as suboxides (TiO_{2-x})

Some of the principal chemical reactions involved in our experiments can be summarized as following: $\text{TiCl}_4 + \text{O}_2 \rightarrow \text{TiO}_2 + 2 \text{Cl}_2$ and $\text{C}_6\text{H}_{18}\text{Si}_2\text{O} + 12 \text{O}_2 \rightarrow 2 \text{SiO}_2 + 6 \text{CO}_2 + 9 \text{H}_2\text{O}$

$\text{C}_2\text{H}_4 + 3 \text{O}_2 \rightarrow 2 \text{CO}_2 + 2 \text{H}_2\text{O}$ for stoichiometric amount of oxygen; As discussed, we must account for the reactions in an insufficient oxidant conditions:

$\text{C}_2\text{H}_4 + 2 \text{O}_2 \rightarrow 2 \text{CO} + 2 \text{H}_2\text{O}$ and $\text{C}_2\text{H}_4 + \text{O}_2 \rightarrow 2 \text{CO} + 2 \text{H}_2$ or even in the absence of O_2 :

$\text{C}_2\text{H}_4 \rightarrow \text{C}_2\text{H}_2 + \text{H}_2$; $\text{C}_2\text{H}_2 \rightarrow 2 \text{C} + \text{H}_2$; $\text{C}_2\text{H}_4 \rightarrow 2 \text{C} + 2 \text{H}_2$ and $\text{C}_6\text{H}_{18}\text{Si}_2\text{O} \rightarrow (\text{C}_x\text{H}_y\text{Si}_z\text{O}_w) + n \text{CH}_4 + m \text{H}_2$

The amount of vapors of Ti and Si precursors depend of the pressure in the bubblers (approximated as being the same with the working pressure in the reaction chamber) and on the carrying gas flows through the volatile liquids (ethylene/air).

In the TLOP8 experiment a tronconical deposit was also formed on the central nozzle rim. This deposit was also collected and shows a silicon and titanium segregation (by EDX elemental evaluation). This can be explained by the incomplete mixing of TIC14 flow from the center and HMDSO from the annular co-flow. SEM image (fig4) also show the more compact morphology from this special deposit. The fine morphology of the resulted powders is visible also in the other SEM images from fig. 4 and 5.

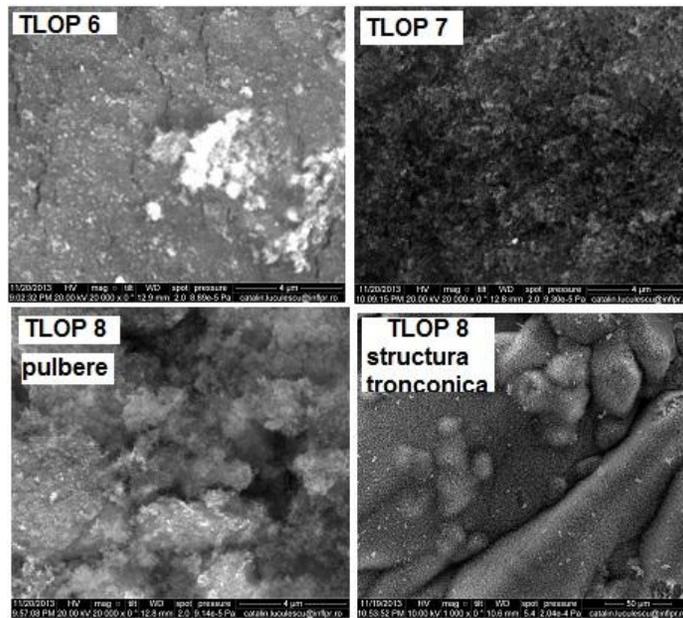


Fig.4 SEM images from nanopowders resulted in TLOP 6,7 and 8 experiments

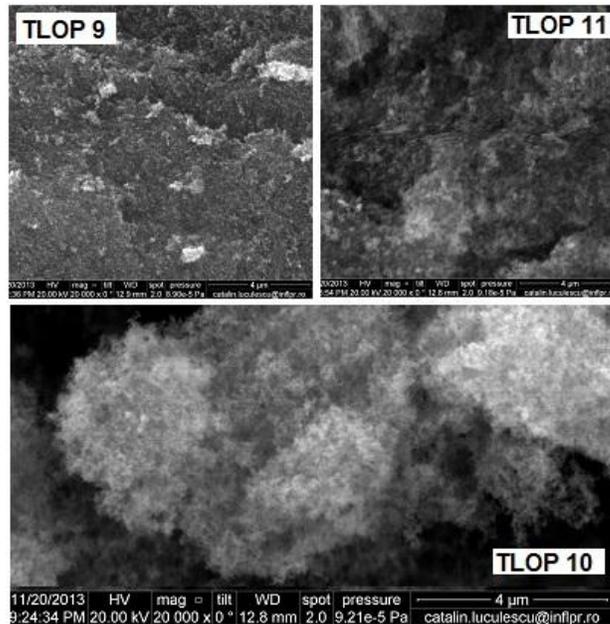


Fig.5. SEM images from nanopowders resulted in TLOP 9,10 and 11 experiments

The TLOP11 SEM image (fig.5, down) witnesses also the nanometric, porous morphology specific to the laser pyrolysis synthesized powders

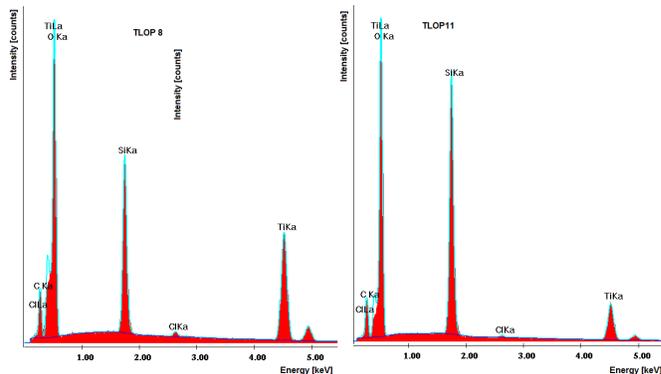


Fig.6 EDX spectra from TLOP8 and 11 nanopowders

Sample	Ti [at%]	Si [at%]	O [at%]	C [at%]	Cl [at%]	Ti/Si atomic ratio
TLOP 01	21.23	-	59.68	18.85	0.24	-
TLOP 04	14.61	-	60.72	24.31	0.36	-
TLOP5	19.46	5.80	47.38	27.21	0.15	3.35
TLOP6	14.24	5.99	54.95	24.68	0.14	2.38
TLOP7	16.95	8.92	54.12	19.86	0.14	1.90
TLOP8	19.22	9.79	55.54	15.12	0.33	1.96
TLOP9	15.16	7.77	44.60	32.18	0.29	1.95
TLOP10	8.27	11.80	37.19	42.28	0.46	0.7
TLOP11	9.09	18.08	52.62	20	0.21	0.5

Table 2. Elemental composition (atomic percents) from TLOP samples by EDX analyses

The EDX analyses can't discriminate the bonding and phases but can allow an estimation of the elemental percent of Ti, S, C and O atoms (see Table 2). Generally, the low percent of air introduced results in powders with a higher carbon content as expected. Also the ratio between titanium and silicon atoms can be related with the ratio of the precursors, knowing that from one mole of $TiCl_4$ results one mole of titania whereas from one mole of HMDSO results two moles of silica. Small amount of chlorine is retaining in the raw powders from the $TiCl_4$ precursor. TLOP10 sample resulting from the most oxygen-deficient mixture (minimum amount of air from all the experiments) has the highest atomic percent of carbon = around 40%

The presence of anatase and rutile titania phases is visible in the X-ray diffractograms from fig. 7 and 8. A very broad peak ($2\theta=20..26^\circ$) visible as a baseline bump in TLOP10 sample is due to the amorphous/disordered carbon mixed with amorphous silica. The TLOP11 sample presents a similar yet smaller peak - in this case the amorphous silica seems to be the main source

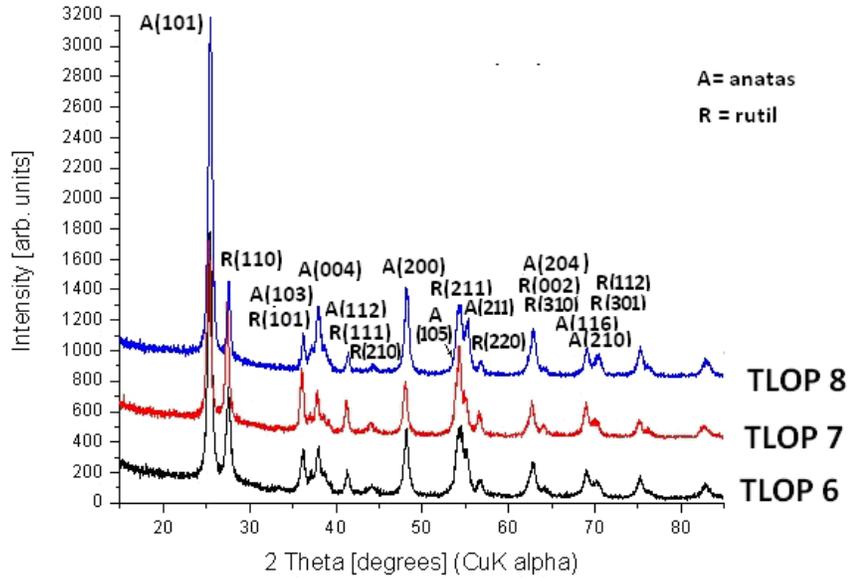


Fig.7 X-ray Superposed diffractograms of eaw TLOP 6, 7 si 8 nanopowders

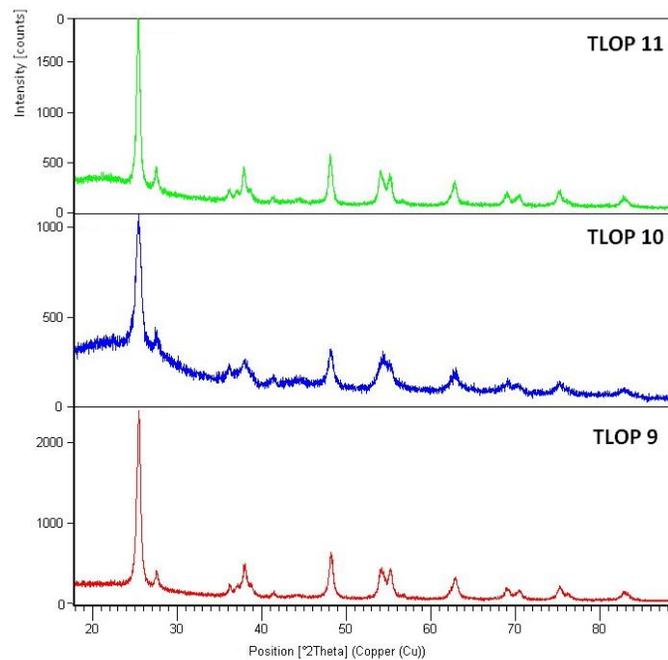


Fig.8 X-ray Superposed diffractograms of raw TLOP 9, 10 and 11 nanopowders

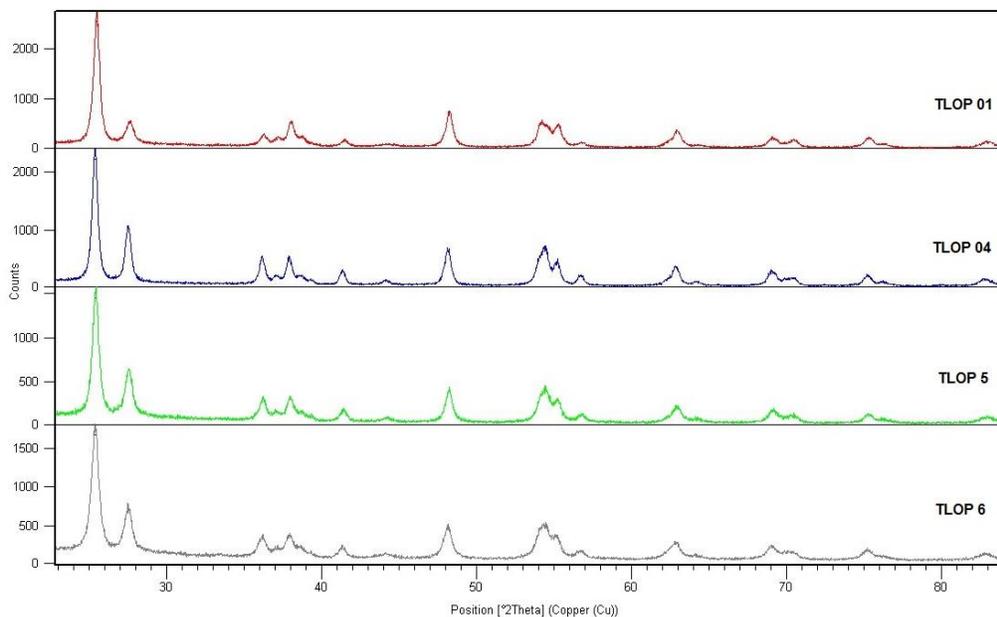


Fig.9 X-ray Superposed diffractograms of raw **TLOP 01, 04, 5 si 6** samples

No signature of titanium suboxides can be seen in the diffractograms from fig. 7, 8 and 9. It is very possible that these compounds to be amorphous and to be integrated in the background noise

Sample	Anatase crystallite [nm]	mean size	Rutile crystallite [nm]	mean size	%Anatas/ /[%Anatas + %Rutil]	%Rutil/ /[%Anatas + %Rutil]
TLOP 01	22		17		83.3	17.7
TLOP 04	24		23		65.8	34.2
TLOP 5	19		15		69.04	30.96
TLOP 6	17		15		39.36	60.64
TLOP 7	18		19		53.55	46.45
TLOP 8	18		24		78	22
TLOP 9	15		15		90.52	9.48
TLOP 10	11		6		85.06	14.94
TLOP 11	21		18		86.69	13.31

Table 3. Crystallite meand sized (anatase and rutile) and their ratio in TLOP nanopowders

A very broad peak centered around 20~26° can also be found in various disordered carbonaceous materials such as those obtained from incomplete (incipient) carbonization of organic substances

such as saccharose dehydration/carbonization with concentrated sulfuric acid or from pyrolysis of various wooden products [20], [21]. As can be seen from the Table 3, the samples obtained when HMDSO was introduced through the central nozzle show the higher anatase content 85-90% .

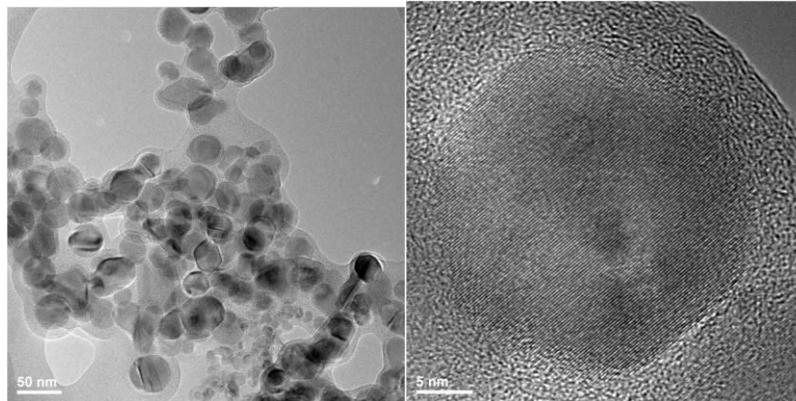


Fig.10 TEM image (left)and HRTEM (right) from TLOP6 sample

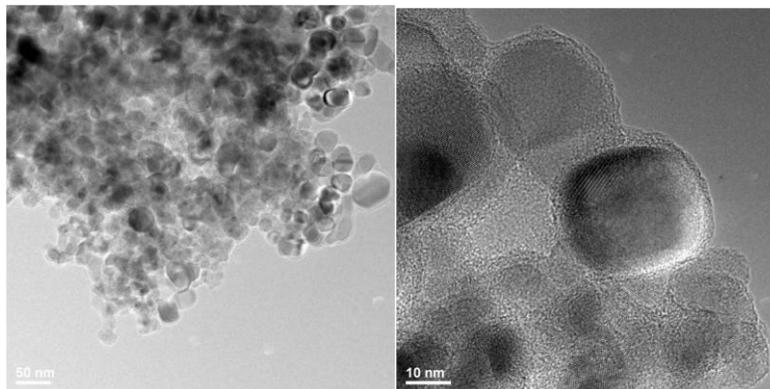


Fig.11 TEM image (left)and HRTEM (right) from TLOP7 sample

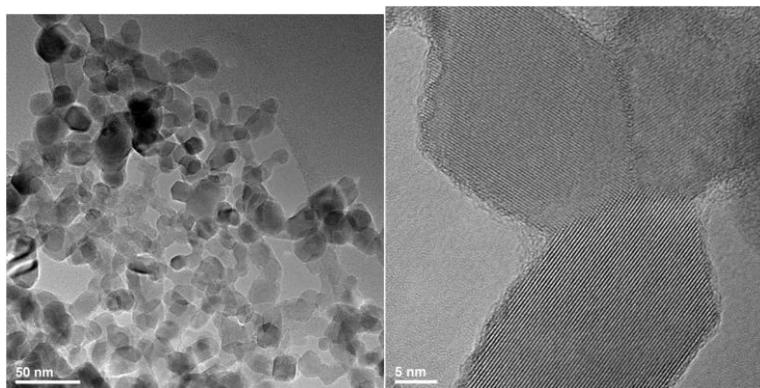


Fig.12 TEM image (left)and HRTEM (right) from TLOP8 sample

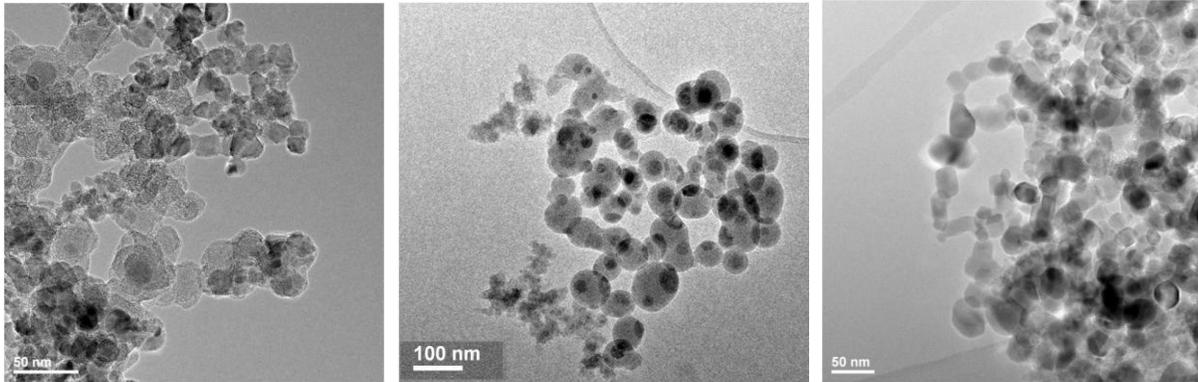


Fig.13 TEM Images from TLPO 9(left), TLOP10 (center) and TLOP11 (right) nanopowders

TEM images from fig.10-13 show chained/aggregated nanoparticles some with a core-shell morphology. Higher magnification TEM from fig 10 and 11 clearly show a crystalline nanoparticles surrounded by a disordered/amorphous shell that must consist of carbon and/or silica. Generally the size of nanoparticles are under 100 nm, with many having sizes around 10-30 nm, depending of sample.

Generally, the UV-VIS absorbance spectra from the ultrasonicated diluted suspensions from the raw TLOP nanopowders (presented in fig.14 and 13) show the typical high absorption behavior in ultraviolet zone, extending for some to the 400-500 nm visible region

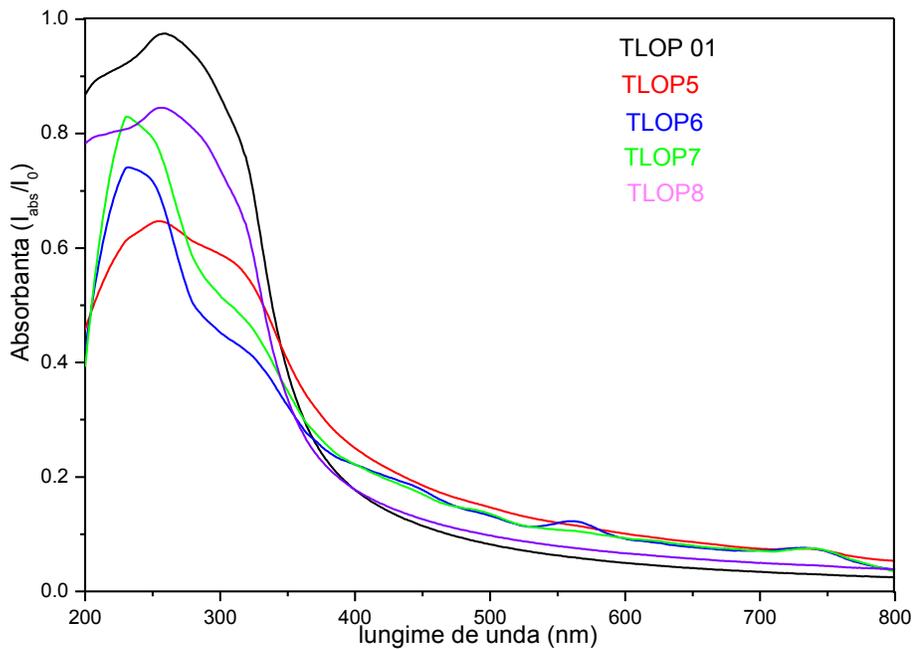


Fig.14 UV-VIZ superposed spectra from TLOP01, si TLOP 4, 5, 6, 7 and 8 nanoparticles as aqueous ultrasonicated suspensions

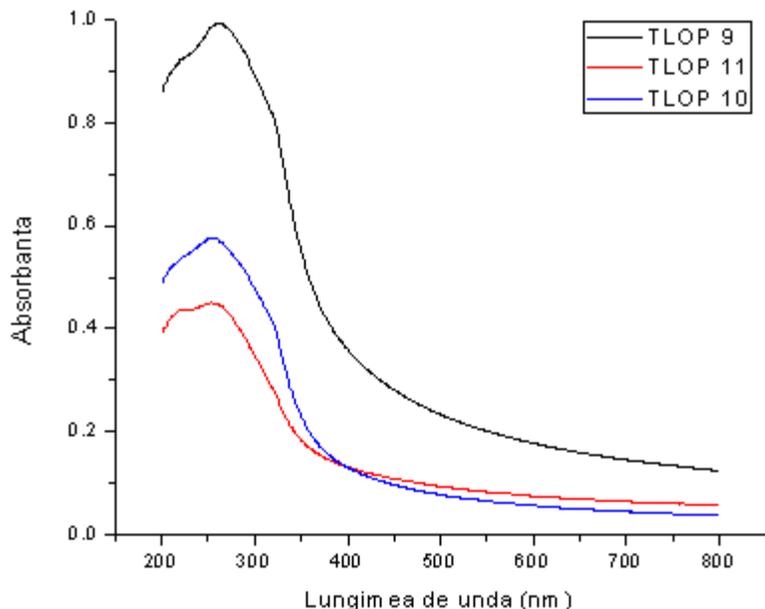


Fig.15 UV-VIZ superposed spectra from TLOP01, si TLOP 9,10 and 11 nanoparticles as aqueous ultrasonicated suspensions

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Project Director, Dr. Claudiu Fleaca

